

Al-Fe-Ni (Aluminum-Iron-Nickel)

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The previous review of this system by [1988Ray] presented liquidus and solidus projections for Al-poor Fe-Ni alloys and for compositions near the Al corner, a reaction sequence for the solidification reactions, isothermal sections at 1250, 950, 850, and 750 °C for Al-poor compositions, and a section at 600 °C near the Al corner. An update by [1994Rag] presented isothermal sections at 1050 and 950 °C from [1982Kha] and a vertical section along the Ni₃Al-Ni₃Fe join from [1987Mas]. A number of new reports on the phase equilibria of this system have since appeared.

Binary Systems

The Al-Fe phase diagram [1993Kat] shows that the face-centered cubic (fcc) solid solution based on Fe is restricted by a γ loop. The body-centered cubic (bcc) solid solution exists in the disordered A2 form (α), as well as the ordered B2 and DO₃ forms. Apart from the high-temperature phase ϵ , there are three other intermediate phases in this system: FeAl₂ (triclinic), Fe₂Al₅ (70-73 at.% Al, orthorhombic), and FeAl₃ or Fe₄Al₁₃ (74.5-76.6 at.% Al, monoclinic). The Al-Ni phase diagram [1993Oka] depicts five intermediate phases: Ni₃Al (23-28 at.% Al, AuCu₃-type cubic, denoted γ'), Ni₅Al₃ (32-36 at.% Al, Ga₃Pt₅-type orthorhombic), NiAl (31-58 at.% Al, CsCl-type cubic), Ni₂Al₃ (58-63 at.% Al, D5₁₃ type hexagonal); and NiAl₃ (75 at.% Al, Fe₃C-type orthorhombic). Recently, [2001Miu] reinvestigated the liquidus in the Ni-rich region of the Al-Ni diagram. [2002Bit] reassessed the solidus of the AlNi (B2) phase in this binary system. The Fe-Ni phase diagram [1993Swa] is characterized by a very narrow solidification range with a peritectic reaction at 1514 °C, between bcc δ and liquid, that yields the Fe-based fcc solid solution. A continuous solid solution denoted γ between fcc Fe and Ni is stable over a wide range of temperatures. At 517 °C, an ordered phase FeNi₃ forms congruently from γ .

Ternary Phases

[1988Ray] listed two ternary compounds in this system: FeNiAl₉ (designated τ_1 , monoclinic) and Fe₃NiAl₁₀ (τ_2 , hexagonal). At the composition FeNiAl₅, [1990Ell] found a phase of hexagonal symmetry (*hP*28, *a* = 0.7703 nm and *c* = 0.7668 nm), which is isomorphous with Co₂Al₅. This phase could be the same as τ_2 in [1988Ray].

[1982Kha] found a new ternary phase of unknown structure (denoted as Q here) at the composition Fe_{5.4}Ni₂₃Al_{71.6}, which was reported to exist in a narrow temperature range. [1996Lem] found a decagonal quasi-crystalline phase around this composition. It is understood here that the two phases refer to the same quasi-crystalline phase [2004Gru].

According to [1996Lem], this phase forms at 930 °C through a ternary peritectic reaction (FeAl₃ + Ni₂Al₃ + L \leftrightarrow Q) and decomposes through a ternary eutectoid reaction at 847 °C (Q \leftrightarrow Ni₂Al₃ + NiAl₃ + FeAl₃). It may be noted that [1982Kha] reported the peritectic formation of Q at 925 °C and did not present an isothermal section at 900 °C, pending confirmation of the stable presence of Q.

Isothermal Sections

[1994Jia] studied the partitioning of alloying elements among the γ , γ' , and B2 phases in Ni-Al base systems. Diffusion couples were prepared from induction-melted alloys and were annealed between 1300 and 800 °C for 10 to 1000 h. The composition of the coexisting phases was measured by electron probe microanalysis and listed. In the equilibrium of γ'/γ , Fe segregates preferentially in the γ phase. The partition coefficient K_{Fe} for γ'/γ equilibrium decreases from 0.75 at 1300 °C to \sim 0.33 at 800 °C. In the equilibrium of $\gamma'/B2$, Fe segregates preferentially in the B2 phase. Here, K_{Fe} decreases from \sim 0.9 at 1300 °C to \sim 0.75 at 900 °C. [1994Jia] constructed partial isothermal sections at 1300 and 1100 °C. The section at 1100 °C is redrawn in Fig. 1.

Using starting metals of 99.99% Al, \sim 99.9% Fe, and electrolytic Ni, [1996Kal] arc-melted alloys with Al \leq 50 at.%, which were annealed at 1127 °C for 100 h. The phase

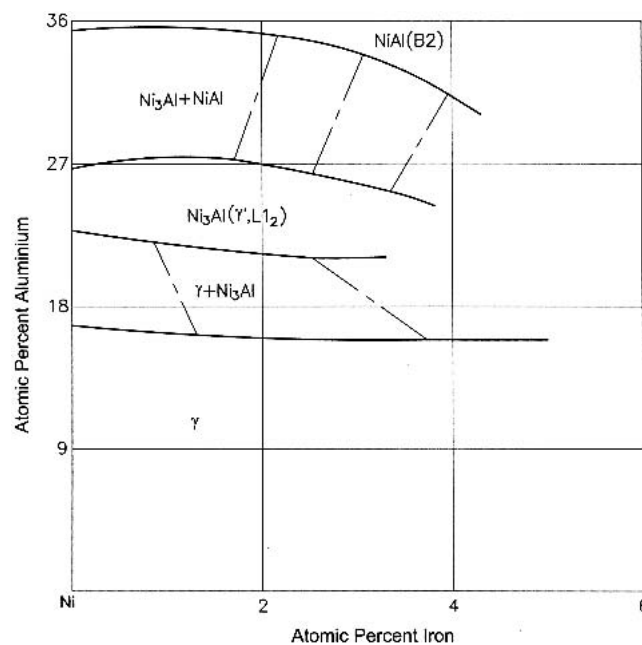


Fig. 1 Al-Fe-Ni partial isothermal section at 1100 °C [1994Jia]

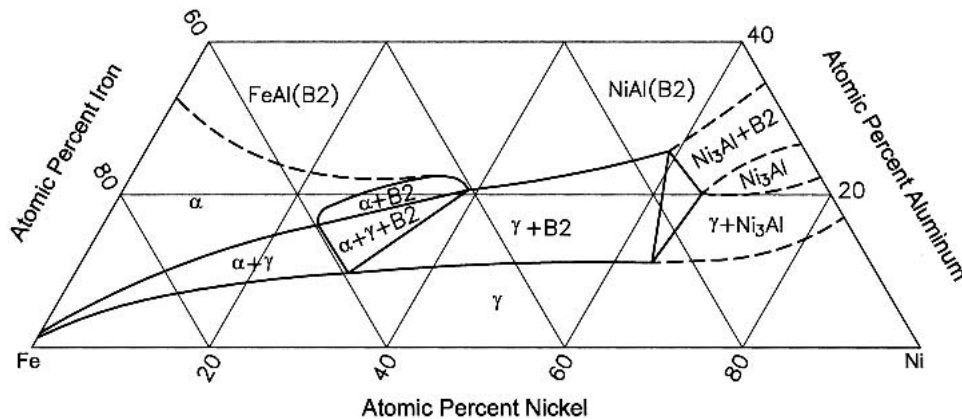


Fig. 2 Al-Fe-Ni isothermal section at 1127 °C [1996Kal]

equilibria were studied by the electron probe microanalysis and x-ray powder diffraction technique. The isothermal section constructed at 1127 °C by [1996Kal] is redrawn in Fig. 2. A small two-phase region of (B2 + α) is located in the composition ranges of Fe and Al of 41 to 61 and 17 to 24 at.%, respectively. The second-order $\alpha \rightarrow B2$ transition without a two-phase boundary extends from the Fe-Al side into the ternary region up to ~30 at.% Ni.

Using neutron-diffraction analysis and calorimetric measurements, [1998Gom] constructed a vertical section along the Ni_3Al-Ni_3Fe join. This section is similar to the vertical section reviewed by [1994Rag]. Recently, the solidus temperatures of alloys along the $NiAl-FeAl$ join were redetermined by [2002Bit]. The melting point of the stoichiometric $NiAl$ was found to be 1681 °C, which is 43° higher than the data from the literature. The solidus temperature initially falls steeply at -13 °C/at.% Fe near the $NiAl$ end. Beyond 5 at.% Fe, it changes to a nearly linear decrease of approximately -8.5 °C/at.% Fe.

[1995Shi] made a theoretical analysis of the effect of Fe on the B2 phase stability. [2001Tan] measured the variation of the lattice parameter of the B2 phase in the ternary region and plotted contour lines of constant lattice parameter as a function of composition. [2000Mar] measured the lattice parameter, density, hardness, and magnetic properties of alloys along the Fe-NiAl join.

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